

Electron scattering dependence of dendritic magnetic instability in superconducting MgB₂ films

Z. X. Ye, Qiang Li,^{a)} and Y. F. Hu

Materials Science Department, Brookhaven National Laboratory, Upton, New York 11973

A. V. Pogrebnyakov and Y. Cui

Department of Physics, Pennsylvania State University, University Park, Pennsylvania 16802

X. X. Xi

Department of Physics, Pennsylvania State University, University Park, Pennsylvania 16802

and Department of Materials Science and Engineering, Pennsylvania State University, University Park, Pennsylvania 16802

J. M. Redwing

Department of Materials Science and Engineering, Pennsylvania State University, University Park, Pennsylvania 16802

Qi Li

Department of Physics, Pennsylvania State University, University Park, Pennsylvania 16802

(Received 21 June 2004; accepted 30 September 2004)

Magnetic instability in both ultrapure and carbon-doped MgB₂ films is investigated by magneto-optical imaging, transport, and bulk magnetization measurements. In the carbon-doped MgB₂ thin films, familiar dendritic flux-jump patterns were observed at low temperature as reported in previous experiments. In the ultrapure MgB₂ thin film, however, a remarkably stable flux penetration was observed, clearly showing the classic behavior of the critical state model. Such different behavior indicates that the electron scattering ultimately controls the magnetic stability of the MgB₂ films. © 2004 American Institute of Physics. [DOI: 10.1063/1.1827931]

Potential application of superconducting MgB₂ in power transmission cables, magnets, motors must take the magnetic stability of the conductor into account. A critical issue is stability against possible flux jumps. This is an avalanche process where flux motion dissipates heat and leads to a local temperature rise which reduces local pinning and facilitates further flux motion.^{1,2} Unprotected flux jumps can sometimes result in a thermal runaway and destroy superconducting equipment. Strong flux instabilities in thin films and bulk MgB₂ at low temperatures have been reported.³⁻⁶ Using magneto-optical imaging (MOI) techniques, Johansen *et al.* revealed that below 10 K the penetration of magnetic flux in pulsed-laser-deposited (PLD) MgB₂ films was dominated by dendritic structures abruptly formed in response to an applied field.³ Though dendritic magnetic instability is believed to be of thermomagnetic origin, the phenomenon is still poorly understood. Recently, Johansen *et al.*³ and Aranson *et al.*⁷ numerically simulated dendritic flux jump in type II superconductors based on the thermal feedback mechanism. However, these simulations are unable to relate the occurrence of flux jump quantitatively to the parameters specific to a particular superconductor, such as its critical temperature T_c , upper critical field H_{c2} , normal state resistivity ρ_n , and critical current density J_c , etc. Furthermore, the central issue of what controls the magnetic stability, particularly in MgB₂, remains unsolved.

The goal of the present work is to explore suppression of the magnetic instability in MgB₂ by varying materials properties, and to determine the key factor responsible for the

dendritic flux jumps in MgB₂ films. Experiments focused on high quality MgB₂ thin films made by hybrid physical-chemical vapor deposition (HPCVD).⁸ It has been shown previously that the HPCVD process is very successful in producing ultrapure MgB₂ films with very low resistivity and clean-limit behavior (“ultra-pure”), being much cleaner than even pure films made by PLD or other route.⁹ In addition, films with a broad range of “dirty-limit properties” could also be obtained through carbon doping via HPCVD process.¹⁰ Magnetic behaviors of these films were studied by a combination of MOI, transport, and bulk magnetization measurements. These results were then compared to our earlier assessments of pure MgB₂ films made by PLD.⁵ Our studies showed that dendritic flux jumps can be completely eliminated by keeping ρ_n sufficiently low, such as in the ultrapure MgB₂ thin films made by HPCVD.

Two types of MgB₂ film (ultrapure and C-doped) were grown on *c*-cut SiC single crystalline substrates using the *in situ* HPCVD process described previously.⁸ The ultrapure films are clean and *epitaxial* with the *c*-axis perpendicular to the surface, while the C-doped films are uniaxially oriented with the *c*-axis perpendicular to the surface, similar to most of the PLD films reported so far. Films sized 5 × 5 mm² of ultrapure MgB₂ (330 nm thick) and 0.12 formula carbon doped MgB₂ (200 nm thick) were selected for MOI study. T_c are 41.2 and 38.4 K for the ultrapure and the C-doped samples, respectively. Flux motion in these MgB₂ thin films was directly recorded using the high resolution MOI station described elsewhere.⁵ Bulk magnetization was measured on the *same* films in a Quantum Design MPMS SQUID magnetometer with the applied field $H_a \perp$ film surface.

^{a)} Author to whom correspondence should be addressed; electronic mail: qiangli@bnl.gov

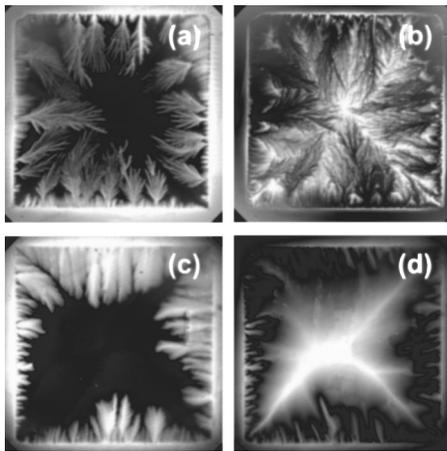


FIG. 1. Magneto-optical images showing flux penetration patterns in the zero-field-cooled C-doped and ultrapure MgB₂ films (5 × 5 mm²) at 4.2 K: (a) dendritic flux penetration into the C-doped film at $\mu_0 H_a = 8$ mT; (b) remanent state of the C-doped film showing the exit of the dendritic flux after $\mu_0 H_a$ reduced from 0.1 T to zero; (c) stable and gradual flux penetration into the ultrapure film at $\mu_0 H_a = 20$ mT. Note: at $\mu_0 H_a < 10$ mT, nearly perfect shielding (no flux entry) was observed; (d) remanent state of the ultrapure film showing regular roof-top flux trapping pattern after $\mu_0 H_a$ reduced from 0.1 T to zero.

Figure 1 shows the MO images. Striking differences in the flux penetration patterns are immediately apparent between C-doped [(a) and (b)] and ultrapure MgB₂ films [(c) and (d)]. These images were taken at 4.2 K during the initial magnetization (a) and (c) after zero-field cooling and in the remanent states (b) and (d). Figure 1(a) shows vigorous dendritic flux penetration into the C-doped film at external field $\mu_0 H_a = 8$ mT. The dendrites nucleate randomly near the film edge and propagate into the films immediately. Figure 1(b) shows the remanent state of the C-doped film after $\mu_0 H_a$ was reduced from 0.1 T to zero, where the dark dendrites show the sudden exit of flux. In general, the behavior of the dendritic flux motion in these C-doped films is essentially the same as that observed previously in pure MgB₂ films made by PLD.^{3,5} Similarly, at $T > 10$ K, MOI did not reveal any flux jumps in the C-doped films. In a striking contrast, dendritic flux jumps are completely absent in the ultrapure MgB₂ films. Instead, a regular and gradual flux penetration was observed, as shown in Fig. 1(c) taken at $\mu_0 H_a = 20$ mT and in Fig. 1(d) taken after $\mu_0 H_a$ was reduced from 0.1 T to zero, being consistent with the prediction of the critical state model. The MO images of Figs. 1(c) and 1(d) are not indicative of completely homogenous flux penetration, as can be seen in some other superconducting systems.

The magnetic behavior of the C-doped and the ultrapure MgB₂ films was further explored by bulk magnetization measurements. Figure 2 shows that the initial magnetization of the C-doped film contains small flux jumps, as indicated by pronounced noise illustrated in the inset to Fig. 2. Such behavior was observed at $1.8 \text{ K} \leq T \leq 9.5 \text{ K}$ for $\mu_0 H_a$ up to 0.15 T. At $T \geq 10$ K, both initial and full hysteresis measurements gave smooth magnetization curves for the C-doped film. In contrast, smooth magnetization curves were always observed at all temperatures, as low as 1.8 K, for the ultrapure film. These observations are in excellent agreement with those flux profiles found in the MOI studies above.

To determine the key factors responsible for the disappearance of the dendritic flux jumps in the ultrapure MgB₂

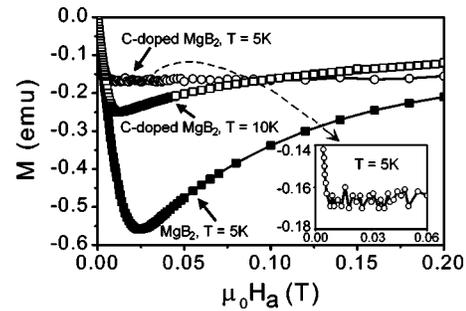


FIG. 2. Initial magnetization of the C-doped and ultrapure MgB₂ films. The inset is an expanded view of the magnetization curve for the C-doped film at 5 K, showing pronounced noise due to flux jumps. A regular magnetization behavior was observed for the ultrapure MgB₂ film at all temperatures from 1.8 K to T_c , as well as for the C-doped film at $T \geq 10$ K.

films, we analyze various parameters relevant to the thermo-magnetic instability. We first examine $J_c(T)$. In general, the higher the value of J_c is, the steeper the slope of the flux gradient in the critical state is, and hence the higher is the tendency for a breakdown of a critical state with respect to hot spot formation. In addition, higher $|dJ_c/dT|$ gives faster propagation of this breakdown. Figure 3 shows the T dependence of J_c obtained for the C-doped and the ultrapure films from both transport measurements of similarly processed films in self field,¹¹ and magnetization measurements in remanent field of the actual films viewed by MOI (the standard Bean model is applied to magnetic hysteresis). The large error bar in J_c for the C-doped film at 4.2 K is due to slight differences in C-concentration among the films. In fact, the ultrapure film has higher J_c and $|dJ_c/dT|$ than the C-doped film. This rules out the possibility that J_c is responsible for the absence of dendritic flux jumps in the ultrapure MgB₂ films.

Next, we examine the conditions required for local flux jumps, which depend on the ratio τ of the flux (t_m) and thermal (t_t) diffusion time constants.^{1,2,7} The dimensionless parameter τ is given by $\tau = t_m/t_t = D_t/D_m = \mu_0 \kappa / C \rho_f$, where the magnetic diffusivity $D_m = \rho_f / \mu_0$, thermal diffusivity $D_t = \kappa / C$, μ_0 is the permeability of vacuum; and κ , C , and ρ_f are the superconductor's thermal conductivity, heat capacity, and flux flow resistivity, respectively. Under local adiabatic conditions, where $\tau \ll 1$ ($t_m \ll t_t$), the magnetic flux diffusion is considerably faster than that of heat, and there is not enough time to redistribute and remove the heat released due to flux motion. Under these conditions, dendritic flux jumps occur. Such conditions applied to earlier experiments on thin

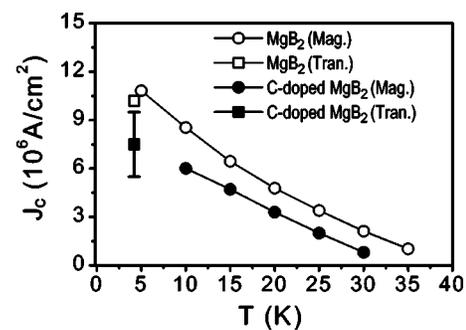


FIG. 3. Temperature dependence of J_c for the C-doped and ultrapure MgB₂ films determined by the transport (Tran.) and magnetization (Mag.) methods.

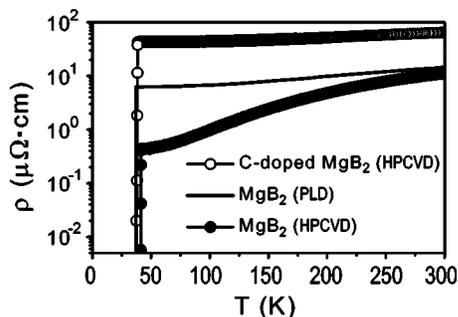


FIG. 4. Temperature dependence of resistivity of the C-doped and ultrapure MgB₂ films plotted along with the pure MgB₂ film made by PLD (Refs. 5 and 16).

films of Nb (Ref.12) and MgB₂ as discussed earlier.^{3,5} Although it is not feasible for us to measure the thermal conductivity and the heat capacity of these MgB₂ films directly, other works on the C-doped MgB₂ samples suggest that these parameters should not be too different from those of pure MgB₂.¹³ It becomes clear that the key parameter that changes the local adiabatic condition is the flux flow resistivity ρ_f . For simplicity, ρ_f is approximated by $\rho_f = \rho_n H_a / H_{c2}$.¹⁴ It is clear that ρ_f must be greatly reduced in order to slow down the flux diffusion. This can be accomplished by either a drastic reduction of ρ_n or a large increase of H_{c2} . In one-band superconductors these trends oppose each other, H_{c2} being proportional to ρ_n , and thus it is not possible to separately alter ρ_n and H_{c2} easily.¹⁴ However, in two-band superconductors it is possible to embody strong electron scattering and high H_{c2} at low temperature in one dirty band while keeping low ρ_n in the other clean band.¹⁵

Figure 4 shows the T dependence of resistivity for the ultrapure and C-doped MgB₂ films at zero field, plotted along with the result for the PLD film reported earlier.^{5,16} T_c is similar for all three films (~ 38 – 41 K), while the difference in ρ_n at T_c is huge (~ 0.4 $\mu\Omega$ cm for the ultrapure and ~ 40 $\mu\Omega$ cm for the C-doped film, respectively, while for the earlier PLD film, ρ_n at T_c is ~ 7 $\mu\Omega$ cm). This dramatic change in resistivity (over two orders of magnitude) is not compensated for by threefold changes in H_{c2} . $\mu_0 H_{c2}$ for the C-doped film was found to be ~ 25 T for $H//c$ at 4.2 K, based on the C-doping, as compared to 7 T in the ultrapure MgB₂ films.¹⁰ Taking $H_a = 0.1$ T as an example, ρ_f for the ultrapure MgB₂ film is ~ 0.006 $\mu\Omega$ cm, versus 0.16 $\mu\Omega$ cm in our carbon doped sample. For the earlier pure MgB₂ film made by PLD,^{5,16} ρ_f is ~ 0.1 $\mu\Omega$ cm, where $H_{c2} \sim 7$ T at 4.2 K was used. In fact, our C-doped film and earlier PLD films have comparable ρ_f at low temperatures. Thus, it is not surprising to observe similar dendritic flux jumps in those films. In comparison, ρ_f in the ultrapure MgB₂ films is about two orders of magnitude lower. This drastically reduced ρ_f in the ultrapure MgB₂ film slows flux diffusion considerably, and hence prevents dendritic flux jumps.

In summary, we report the absence of dendritic flux jumps in the ultrapure MgB₂ films at temperatures as low as 1.8 K. Similar dendritic flux jump behavior observed in C-doped MgB₂ films and earlier pure MgB₂ films made by PLD is likely due to their comparable values of flux flow resistivity. It is shown that the magnetic stability in MgB₂ films is closely related to their normal state resistivity. The onset of dendritic instabilities in dirty-limit MgB₂ films perhaps manifests another unique feature of electron scattering in the two-band superconductors.

The authors would like to thank Dr. L. Cooley for critical reading of this manuscript. The work at BNL was supported by the U.S. Department of Energy, Office of Basic Energy Science, under Contract No. DE-AC-02-98CH10886. The work at Penn State is supported in part by ONR under Grant Nos. N00014-00-1-0294 (X.X.X.) and N0014-01-1-0006 (Redwing), by NSF under Grant Nos. DMR-0306746 (X.X.X. and J.M.R.), DMR-0405502 (Q.L.).

¹R. G. Mints and A. L. Rakhmanov, Rev. Mod. Phys. **53**, 551 (1981), and references therein.

²S. L. Wipf, Cryogenics **31**, 936 (1991).

³T. H. Johansen, M. Baziljevich, D. V. Shantsev, P. E. Goa, Y. M. Galperin, W. N. Kang, H. J. Kim, E. M. Choi, M.-S. Kim, and S. I. Lee, Europhys. Lett. **59**, 599 (2002); M. Baziljevich, A. V. Bobyl, D. V. Shantsev, E. Altshuler, T. H. Johansen, and S. I. Lee, Physica C **369**, 93 (2002).

⁴Z. W. Zhao, S. L. Li, Y. M. Ni, H. P. Yang, Z. Y. Liu, H. H. Wen, W. N. Kang, H. J. Kim, E. M. Choi, and S. I. Lee, Phys. Rev. B **65**, 064512 (2002).

⁵Z. Ye, Q. Li, G. D. Gu, J. J. Tu, W. N. Kang, E.-M. Choi, H.-J. Kim, and S.-I. Lee, IEEE Trans. Appl. Supercond. **13**, 3722 (2003); Q. Li, G. D. Gu, and Y. Zhu, Appl. Phys. Lett. **82**, 2103 (2003); Q. Li, L. Wu, Y. Zhu, A. R. Moodenbaugh, G. D. Gu, M. Suenaga, Z. X. Ye, and D. A. Fischer, IEEE Trans. Appl. Supercond. **13**, 3051 (2003).

⁶S. Jin, H. Mavoori, C. Bower, and R. B. van Dover, Nature (London) **411**, 563 (2001).

⁷I. Aranson, A. Gurevich, and V. Vinokur, Phys. Rev. Lett. **87**, 067003 (2001).

⁸X. H. Zeng, A. V. Pogrebnikov, A. Kotcharov, J. E. Jones, X. X. Xi, E. M. Lysczek, J. M. Redwing, S. Y. Xu, Q. Li, J. Lettieri, D. G. Schlom, W. Tian, X. Q. Pan, and Z. K. Liu, Nat. Mater. **1**, 35 (2002).

⁹A. V. Pogrebnikov, J. M. Redwing, J. E. Jones, X. X. Xi, S. Y. Xu, Q. Li, V. Vaithyanathan, and D. G. Schlom, Appl. Phys. Lett. **82**, 4319 (2003); M. Iavarone, G. Karapetrov, A. E. Koshelev, W. K. Kwok, G. W. Crabtree, D. G. Hinks, W. N. Kang, E.-M. Choi, H. J. Kim, H.-J. Kim, and S.-I. Lee, Phys. Rev. Lett. **89**, 187002 (2002).

¹⁰A. V. Pogrebnikov, X. X. Xi, J. M. Redwing, V. Vaithyanathan, D. G. Schlom, A. Soukiassian, S. B. Mi, C. L. Jia, J. E. Giencke, C. B. Eom, J. Chen, Y. F. Hu, Y. Cui, and Q. Li, Appl. Phys. Lett. **85**, 2017 (2004).

¹¹S. Y. Xu, Q. Li, E. Wertz, Y. F. Hu, A. V. Pogrebnikov, X. H. Zeng, X. X. Xi, and J. M. Redwing, Phys. Rev. B **68**, 224501 (2003).

¹²C. A. Durán, P. L. Gammel, R. E. Miller, and D. J. Bishop, Phys. Rev. B **52**, 75 (1995).

¹³R. A. Ribeiro, S. L. Bud'ko, C. Petrovic, and P. C. Canfield, Physica C **384**, 227 (2003), and references therein.

¹⁴P. G. de Gennes, Phys.: Condens. Matter **3**, 79 (1964); N. R. Werthamer, E. Helfand, and P. C. Hohenberg, Phys. Rev. **147**, 295 (1966); K. Maki, *ibid.* **148**, 362 (1966).

¹⁵A. Gurevich, Phys. Rev. B **67**, 184515 (2003).

¹⁶H.-J. Kim, W. N. Kang, E.-M. Choi, M.-S. Kim, K. H. P. Kim, and S.-I. Lee, Phys. Rev. Lett. **87**, 087002 (2001).